THE $PbO_2-PbO-V_2O_5$ **SYSTEM IN AIR**

Y.M. KASSEM, A.N. MAHDY and M.F. ABADIR

Chemical Engineering Department, College of Engineering, Cairo University, Giza (Egypt) (Received 17 January 1989)

ABSTRACT

The equilibrium relationship in air of the system $PbO_2-PbO-V_2O_5$ was studied using the thermobalance technique, as well as combined TG-DTA and X-ray analysis. Three compounds were established: 2PbO \cdot V₂O₅, 3PbO \cdot V₂O₅ and 8PbO \cdot V₂O₅, 4PbO \cdot V₂O₅, was, however, not detected. 2PbO \cdot V₂O₅ can be prepared from PbO₂ at a relatively low temperature (420 °C). The X-ray pattern for the compound 8PbO \cdot V₂O₅ is presented and the compatible phases were established.

INTRODUCTION

TABLE 1

The study of the phase equilibria of this system has been motivated by the increased importance of lead vanadate compositions in various fields of applications. Table 1 shows the various applications for glasses containing V_2O_5 and lead oxides. In addition, V_2O_5 has been recently extensively used

Applications of $PbO-V₂O₅$ glasses

with other oxides in the glass industry, particularly in glasses with special electrical properties [9,10].

PREVIOUS WORK

Lead vanadates

Various lead vanadates have been reported. Table 2 summarizes the available data.

The binary system Pb-0

This system had been previously reported by Abadir et al. [17].

The PbO- V_2O_5 system

Amadori [12] has published the phase diagram in Fig. 1 which shows the presence of 3 intermediate compounds melting congruently without evidence

TABLE 2

Lead vanadates

Fig. 1. PbO-V,O, system, published by Amadori [12].

of any solid solubility. While $2PbO \cdot V_2O_5$ melts at $721^{\circ}C_2$, $3PbO \cdot V_2O_5$ melts at 957° C and $8PbO \cdot V_2O_5$ melts at 797° C. The presence of these compounds was recently confirmed by several authors [16,18]. Their preparation from PbO and $V₂O₅$ required relatively high temperatures $(600-750\degree C)$ and long heating periods ranging from 1 day to 1 week. The compound 4 PbO \cdot V₂O₅ was reported by Shimohira et al. [16] and is not shown in Fig. 1.

In the present work, $PbO₂$ was used as the starting material to take into account the effect of lead polyvalency which has been ignored in previous studies.

EXPERIMENTAL TECHNIQUES

NH,VO,, supplied by VEB Laborchemie Apolda, East Germany, was used to prepare V,O,. Figure 2 shows the combined DTA, TG and DTG results obtained on heating pure $NH₄VO₃$ in the derivatograph (a simulta-

Fig. 2. Thermal analysis of NH₄VO₃ on the derivatograph (rate of heating 10 °C min⁻¹).

neous thermal analyser) at a heating rate of 10° C min⁻¹ in a Pt crucible. Conversion to V_2O_5 , demonstrated by constant weight, occurs at about 400 °C. The DTA peak at 700 °C corresponds to the melting of the oxide. In view of these curves, V_2O_5 was prepared by heating NH₄VO₂ at 600^oC to a constant weight.

PbO, (A.R) grade was also supplied by VEB Laborchemie Apolda, and was found to contain 0.7% maximum impurities. PbO, was dried at 120° C prior to weighing out the appropriate amounts of V_1O_5 and PbO₂ and mixing them in an agate mortar. The mixtures were then suspended in the thermobalance and the temperature raised to 120° C until constant weight was obtained, in order to eliminate the moisture. The mixtures were then heated at different temperatures until equilibrium was reached at each temperature in order to determine the change in O/V ratio in each case. Such a technique has been extensively used to establish non-condensed systems [17,19,20].

EXPERIMENTAL RESULTS AND DISCUSSIONS

The dissociation curves for several mixtures were used to draw the ternary isobaric diagram in air shown in Fig. 3. These results were confirmed by occasional DTA and TG analyses, as well as by X-ray diffraction analysis. The interpretation of the different dissociation curves can be made by considering these curves together with the ternary isobaric section, Fig. 3.

Fig. 3. Ternary isobar in air for the $PbO_2-PbO-V_2O_5$ system (molar basis).

Dissociation curves in air

All the dissociation curves show the following common characteristics. Oxygen is lost from about 100 to 278° C at which a vertical isothermal step occurs. Above 278 \degree C, oxygen is progressively lost up to about 345 \degree C at which a second vertical isothermal step is observed. However the extent of oxygen loss due to that step and, consequently, the location of the final compositions obtained on the ternary diagram depend on the composition of the original mixture. Accordingly, the different dissociation curves were divided into 4 groups.

Group 1. Dissociation curues for mixtures containing more than 33% V,O, (molar)

Figure 4a was selected to show the main features observed in these mixtures. It shows the dissociation curve in air for a mixture initially

Fig. 4. Dissociation curve of (a) $PbO_2 - V_2O_5$ mixture in air for PbO_2 : $V_2O_5 = 3:2$ (molar), (b) **PbO₂ -** V_2O_5 **mixture in air for PbO₂ :** $V_2O_5 = 7.5$ **: 2.5 (molar), (c) PbO₂ -** V_2O_5 **mixture in air** for PbO_2 : $V_2O_5 = 4:1$ (molar), and (d) $PbO_2 - V_2O_5$ mixture in air for PbO_2 : $V_2O_5 = 9:1$ **(molar).**

containing 40% V₂O₅. As previously stated, at $278\,^{\circ}$ C a vertical step occurs corresponding to the reaction

$$
\mathrm{PbO}_x(s.s.) \to \mathrm{PbO}_x(s.s.) + (2 - x/2)O_2 \tag{1}
$$

Progressive oxygen loss then takes place until 345° C where a vertical isothermal step takes the O/Pb ratio to about 1.05. On further temperature increase, the ratio approaches 1 at 420" C. As seen from Fig. 3 the last vertical step corresponds to the reaction

$$
2PbO_x(s.s.) + V_2O_5 = 2PbO \cdot V_2O_5(s.s.) + (x-1)O_2
$$
 (2)

The value of x at which this transition occurs was found to be about 1.48 for all mixtures.

Group 2. Dissociation curves for mixtures containing 25-33% $V₂O₅$ *(molar)* Figure 4b shows the dissociation curves obtained on heating a mixture of PbO, and V_2O_5 containing 25% V_2O_5 (molar) under equilibrium conditions in the thermobalance: it shows the main features observed for all mixtures having a V_2O_5 content from 25 to 33%.

As seen from this figure, the step at 278° C is followed by progressive oxygen loss until 345° C at which a vertical step occurs. This step, as previously mentioned, corresponds to the formation of $2PbO \cdot V_2O_5$. However, the O/Pb ratio reached a value of 1.21. As shown in Fig. 3, at this temperature PbO, coexists in equilibrium with $2PbO \cdot V_2O_5$. Progressive oxygen loss then takes place up to about 370° C where a sudden vertical step indicates the formation of Pb_3O_4 according to the reaction

$$
3PbOx \rightarrow Pb3O4 + (3x - 4/2)O2
$$
 (3)

The O/Pb ratio remains constant up to about 425° C, at which a final step occurs shifting the O/Pb ratio to 1. As seen in Fig. 3, this step corresponds to the reaction

$$
2PbO \cdot V_2O_5 + \frac{1}{3}Pb_3O_4 \rightarrow 3PbO \cdot V_2O_5 + \frac{1}{6}O_2 \tag{4}
$$

Evidence of the formation of $3PbO \cdot V_2O_5$ at such low temperatures was obtained by X-ray analysis, as will be discussed later.

Group 3. Dissociation curves for mixtures containing $11-25\%$ V_2O_5 (molar) Dissociation curves of a mixture containing 20% V₂O₅ is shown in Fig. 4c. The O/Pb ratio varies in almost the same way as the 25% V₂O₅ mixture, i.e. the invariant situations observed at 278, 345 and 370° C are present. At the end of the step at 425° C, the O/Pb ratio is 1.09. This ratio remains constant up to about 480° C where a sudden loss changes the O/Pb ratio to unity. This loss in weight corresponds to the reaction

$$
3PbO \cdot V_2O_5 + \frac{5}{3}Pb_3O_4 \rightarrow 8PbO \cdot V_2O_5 + \frac{5}{6}O_2 \tag{5}
$$

Group 4. Dissociation curves for mixtures containing up tb II % V,O, (molar) Figure 4d shows the dissociation curve for a mixture initially containing 10% V₂O₅. The invariant situations are observed once more at the same temperatures of 278, 345, 370, 425 and 480 $^{\circ}$ C, as seen by the corresponding vertical steps. At the end of the step at 480° C, the O/Pb ratio is about 1.1 and remains constant up to 560° C where it drops suddenly to unity. This temperature is the same temperature at which pure Pb_3O_4 transforms to PbO [17] indicating that PbO does not take V_2O_5 into solid solution.

Combined thermal analysis results

Figure 5 shows the combined DTA, TG and DTG results obtained by heating a mixture initially containing 66% PbO, (molar). The thermal curves obtained are as expected in view of the results of the equilibrium dissocia-

Fig. 5. Thermal analysis of $2PbO_2$: V₂O₅ (molar) on the derivatograph (rate of heating 10[°]C min^{-1}).

Fig. 6. Non-stoichiometry of $2PbO \cdot V_2O_5$.

tion curves. In particular, the presence of peak 7 on the DTA which is not accompanied by a change in weight (on the TG and DTG curves), seems to be in favour of the assumed formation of a solid solution of $2PbO \cdot V_2O_5$. If $2PbO \cdot V_2O_5$ was a stoichiometric phase then it would have melted at a definite temperature without any trace of liquid appearing before melting. In that case, we should have obtained one melting peak on the DTG. The fact that peaks 7 and 8 occur without weight change means that melting has occurred over a range of temperatures. This could be the case if $2PbO \cdot V_2O_5$

TABLE 3

Combined thermal analysis of a mixture containing $2PbO_2$: V_2O_5 (molar)

Peak number	Type	Temperature $(^{\circ}\mathrm{C})$	Change in weight	Interpretation
1	endo	140	slight	Loss of moisture
$\overline{2}$	endo	220	slight	Non-stoichiometry of PbO
3	endo	390	yes	Transition to PbO _x
4, 4	endo	$420 - 460$	yes	Non-stoichiometry of $PbOx$
5	endo	520	yes	Transition to $2PbO \cdot V_2O_5$
6	endo	540	slight	Non-stoichiometry of
7	endo	570	no	$2PbO\cdot V_2O_5$ Eutectic formation between
8	endo	720	no	$2PbO\cdot V_2O_5(s.s.)$ and $V_2O_5(s.s.)$ Congruent melting of $2PbO\cdot V_2O_5$

is a non-stoichiometric phase. This situation is shown in Fig. 6. Being performed under non-equilibrium condition (heating rate of 10° C min⁻¹), the combined thermal curves gave transition temperatures higher than the corresponding equilibrium temperatures. Table 3 summarizes the results of the combined thermal analysis.

The ternary isobaric section in air

Figure 3 shows the ternary isobaric section obtained by plotting the beginning and the end of each step obtained on the dissociation curves. The following invariant situations occur.

(i) At 278°C, PbO₂(s.s.), V₂O₅(s.s.) and PbO_x ($x = 1.57$) coexist with oxygen in equilibrium. The composition of the $V_2O_5(s.s.)$ vertex of the tie triangle $V_2O_5-PbO_7-PbO_2$ is not certain. That is why the tie lines joining PbO₂ and V₂O₅(s.s.) as well as the small solid gap at the V₂O₅ vertex are shown dotted in the figure.

(ii) At 345 °C, PbO, $(x = 1.5)$, V₂O₅(s.s.) and 2PbO \cdot V₂O₅(s.s.) coexist in equilibrium with oxygen. The presence of both $V_2O_5(s.s.)$ and $2PbO \cdot$ $V_2O_5(s.s.)$ were confirmed by X-ray analysis as well as by joining compositions reached at the same temperature on the ternary phase diagram (as will be explained when discussing the monovariant situation). The occurrence of this tie triangle implies that the use of $PbO₂$ (instead of PbO) resulted in the formation of $2PbO \cdot V_2O_5$ at 345° C instead of 600° C which was reported when PbO was used $[11]$. This is due to the destabilization of PbO, by reaction with V_2O_5 .

(iii) At 370°C, PbO, $(x = 1.41, 2PbO \cdot V_2O_5(s.s.)$ and Pb₃O₄ coexist in equilibrium with oxygen. This is almost the same temperature at which pure PbO, converts to Pb_3O_4 implying that neither of the two oxides has taken $2PbO \cdot V_2O_5$ into solid solution.

(iv) At 425° C, 2PbO \cdot V₂O₅(s.s.), 3PbO \cdot V₂O₅ and Pb₃O₄ coexist in equilibrium with oxygen. Hence 425° C can be considered as the minimum temperature required for the preparation of $3PbO \cdot V_2O_5$. The formation of $3PbO \cdot V_2O_5$ was confirmed by X-ray analysis.

(v) At $480\degree$ C, $3PbO \cdot V_2O_5$, Pb_3O_4 and $8PbO \cdot V_2O_5$ coexist in equilibrium with oxygen. In view of the absence of X-ray data concerning $8PbO \cdot V_2O_5$, this compound could not be identified by X-ray analysis. The dissociation curve of the mixture initially containing 90% PbO, showed a slight O/Pb drop at 480 °C giving a value of O/Pb at the termination of this step of 1.1. This composition almost lies on a tie line joining Pb_3O_4 and $8PbO \cdot V_2O_5$ suggesting the presence of this latter compound. In view of this result and the work of previous authors, it was placed on the V_2O_5 -PbO edge and the tie line joining this compound to Pb_3O_4 was shown dotted.

(vi) At 560° C, $8PbO \cdot V_2O_5$, Pb_3O_4 and PbO coexist in equilibrium with oxygen. This temperature is identical to the temperature at which the

Fig. 7. Quasi-binary isobaric section in air for the system $PbO₂-PbO-V₂O₅$ (ss, solid solution).

decomposition of Pb_3O_4 to PbO occurs in the case of pure oxides, implying that PbO does not take any V_2O_5 into solid solution.

The following three monovariant situations (where 2 solid phases coexist in equilibrium with oxygen) exist.

(1) At temperatures ranging from about 100 to 278° C, PbO₂ coexists in equilibrium with $V_2O_5(s.s.)$. The tie lines are shown dotted as previously explained.

(2) At temperatures ranging from 278 to 345 °C, PbO, $(1.5 < x < 1.57)$ coexists in equilibrium with $V_2O_5(s.s.)$ and oxygen.

(3) At temperatures ranging from 345 to 370 °C, PbO, $(1.41 < x < 1.5)$ coexists in equilibrium with $2PbO \cdot V_2O_5$ and oxygen. Evidence for the presence of this series of tie lines is obtained from the slight progressive loss in weight observed in the dissociation curves corresponding to compositions containing more than 66% PbO₂ (molar) between 345 and 370 $^{\circ}$ C. When the

O/Pb ratios corresponding to the beginning and end of the progressive weight-loss region on the dissociation curves are projected on the ternary diagram, the points existing at the same temperatures are colinear with PbO_r and $2PbO_x \cdot V_2O_5$ for values of x ranging from 1.5 to 1.41, and for compositions of $2PbO \cdot V_2O_5$ containing a higher proportion of oxygen than the stoichiometric ratio.

Two regions of solid solutions appear on the ternary diagram: $V_2O_5(s.s.)$ the presence of which was confirmed by X-ray diffraction analysis although the extent to which this phase can take lead oxides in solid solution is uncertain; and $2PbO \cdot V, O₅(s.s.)$ which was confirmed by X-ray diffraction. Also, the presence of the solid solution containing excess oxygen was suggested by the location of the tie lines representing $V_2O_5-2PbO \cdot V_2O_5$ and $2PbO \cdot V_2O_5 - PbO$. A projected quasi-binary temperature-composition diagram is shown in Fig. 7.

X-ray *results*

The V,O, solid solution

Identification of a solid solution using X-ray diffraction relies on the slight shift in the d-spacings of the compound which occurs as the solid

TABLE 4

X-ray results for V_2O_5 **and** $V_2O_5(s.s.)$

X-ray diffraction pattern of $2PbO\cdot V_2O_5$

solution is formed [21]. To confirm the presence of V_2O_5 solid solution, a mixture initially containing 95% V_2O_5 and 5% PbO₂ was heated for 48 h at 500 °C. A comparison between the lines obtained for pure V_2O_5 and those of the heated sample is shown in Table 4. As can be seen from this table, there is a slight shift in the values of the d -spacings indicating that up to at least 5% lead oxide can dissolve in V_2O_5 .

The diffraction pattern of $2PbO \cdot V_2O_5$

A mixture initially containing 66% PbO, and 33% V₂O₅ (molar) was heated for 48 h at 510 \degree C and subjected to X-ray analysis. The X-ray pattern of the powder is presented in Table 5 and compared with the pattern reported by Baiocchi et al. [18]. From this table it appears that the lines obtained in the present study are very close to those recently reported in ref. 18, except for slight shifts in d-spacings denoting the formation of the proposed $2PbO \cdot V_2O_5$ solid solution. The presence of this solid solution led Baiocchi et al. [18] to conclude that this phase is richer in V_2O_5 content than $2PbO \cdot V_2O_5$. Five more lines also appeared which were identified as lines characteristic of Pb₃O₄. The presence of Pb₃O₄ is not surprising in view of the proposed formation of $2PbO \cdot V_2O_5$ solid solution rich in V_2O_5 .

TABLE 5

19 4.50	
3.39 8	
2.23 100	
$3.02\,$ 50	
2.99 27	
50 2.90	
2.04 17	
1.90 13	
1.69 14	

Suggested X-ray diffraction pattern of $8PbO\cdot V_2O_5$

TABLE 6

The diffraction pattern of 80% PbO,-20% V,O, (molar)

A mixture initially containing 80% PbO₂-20% V₂O₅ (molar) was heated to 560° C for 48 h and subjected to X-ray analysis. The resulting peaks were compared with the d-spacings of $3PbO \cdot V_2O_5$ [22]. Most of the lines of $3PbO \cdot V_2O_5$ appeared. This suggests that the remaining lines correspond to 8PbO \cdot V₂O₅. Table 6 shows the "screened" lines of 8PbO \cdot V₂O₅.

CONCLUSIONS

From the isobaric sections constructed for the system $PbO_2-PbO-V_2O_5$, the following compounds have been established: $2PbO \cdot V_2O_5(s.s.)$, $3PbO \cdot$ V_2O_5 (P₃V) and 8PbO \cdot V₂O₅ (P₈V) and their stability regions were determined. The previously reported compound $4PbO \cdot V_2O_5$ (P₄V) was not found to exist in air.

The minimum temperature required to obtain P₃V from $PbO_2 + V_2O_5$ is 425 °C and its presence was confirmed by X-ray analysis. X-ray data for P_sV were established.

There are six invariant situations established in air in this system. At the indicated temperatures, three solid phases coexist in equilibrium with oxygen.

(i) At about 278°C, PbO₂(s.s.), V₂O₅(s.s.) and PbO_x (x = 1.57).

(ii) At 345° C, PbO, (x = 1.5), V₂O₅(s.s.) and P₂V.

(iii) At 370°C, PbO_x ($x = 1.41$), P₂V(s.s.) and Pb₃O₄.

(iv) At 425° C, P₂V(s.s.), P₃V and Pb₃O₄.

(v) At 480 $^{\circ}$ C, P₃V, Pb₃O₄ and P₈V.

(vi) At 560° C, P_sV, Pb₃O₄ and PbO.

Also, three monovariant situations occurred where two solid phases and oxygen coexist in equilibrium:

(1) At temperatures ranging from 100 to 278° C, PbO₂(s.s.) coexists in equilibrium with $V_2O_5(s.s.)$ and O_2 .

(2) At temperatures ranging from 278 to 345 °C, PbO, $(1.5 < x < 1.57)$ coexists in equilibrium with V_2O_5 and O_2 .

(3) At temperatures ranging from 345 to 370 °C PbO, $(1.41 < x < 1.5)$ coexists in equilibrium with P_2V and O_2 .

Two regions of solid solutions appear on the ternary diagram:

(a) V_2O_5 (s.s.), the presence of which was confirmed by X-ray identification.

(b) P_yV(s.s.) containing a higher proportion of O_2 than the stoichiometric ratio, the presence of which was also confirmed by X-ray analysis.

REFERENCES

- 1 Tokyo Shibaura Elect. Co. Ltd., Japan, Report 81-26-746 (Cl. CO3C3/30), March 1981.
- 2 E.P. Denton, H. Rauson and J.E. Stanworth, Nature, 173 (1964) 1030.
- 3 A.T. Andreeva, M.N. Verebeichick, P.G. Zhuravleva and D.I. Fridbery, Nov. Neor. Stekl. Matter, 41 (1979).
- 4 S.P. Parker (Ed.), McGraw-Hill Encyclopedia of Science and Technology, Vol. 8, 1987, p. 125.
- 5 Y. Asahara, Hoya Glass Works Ltd., Japan Kokai; 77-69-916 (Cl. CO3C3/16), June 1977.
- 6 V.Z. Petrova, I.V. Filatova, P.E. Kandyba, I. Vazhenin, G.I. Vadov and Yu. Ku. Znetsov, Moscow Inst. of Elect. Technology, U.S.S.R. Patent No. 614-039, July 1978; Chem. Abstr., 89 (1978) 94044e.
- 7 Yu. Goltsov, L.A. Belova, V.P. Cherpillo, V.T. Maltsev and 0.1. Prokopalo, Rostov St. Univ., U.S.S.R. Patent No. 667-530, May 1979; Chem. Abstr., 90 (1979) 82153t.
- 8 V.M. Nogiev, Ah. Fiz. Chim., 52(7) (1978) 174.
- 9 M.M. Ahmed and C.H. Hogarth, J. Mater. Sci., 18(11) (1983) 3305.
- 10 S. Yatabe and S. Kunimoto, Yasmaguchi Daigaku Kogakubu Kenkyu Hokoku, 31(2) (1981) 333; Chem. Abstr. 95 (1981) 106929m.
- 11 T. Tokokawa and O.J. Kleppa, Inorg. Chem., 3(7) (1964) 954.
- 12 M. Amadori, Arm. Inst. Veneto Sd., 76 (11) (1917) 419.
- 13 G. Simon and S.L. Ardelean, Ser. Chem., 15(11) (1970) 43 (in Romanian).
- 14 R.F. Geller and E.N. Bunting, J. Res. Nat. Bur. Stand., 18 (1937) 585.
- 15 I.A. Vasilieva, S.M. Mudrestova and L.V. Stesikova, Ser. Khim., 21(6) (1985) 47 (in Russian).
- 16 T. Shimohira, S. Iwa and H. Tagai, Vozyo Kyokai Shi, 75(868) (1967) 350 (in Japanese).
- 17 M.F. Abadir, A.M. Gadalla and Y.M. Agamawi, J. Trans. Brit. Ceram. Sot., 75(4) (1976) 68.
- 18 E. Baiocchi, M. Bettinelli, A. Monteners and I. Disipio, J. Solid State Chem., 43(l) (1982) 63.
- 19 R. El-Shahat, A.M. Gadalla and J. White, in G. Steward (Ed.), Science of Ceramics, Vol. 3, Academic Presss, London, 1967.
- 20 A.M. Gadalla and M.F. Abadir, Ind. Eng. Chem. Fundam., 23(2) (1984) 220.
- 21 B.D. Cuhity, Elements of X-ray Diffraction, Addison-Wesely, 1959.
- 22 ASTM X-ray Card No. 11-471.